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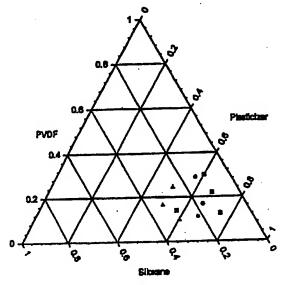
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(54) Title: SINGLE-ION CONDUCTING SOLID POLYMER ELECTROLYTES

(57) Abstract

Novel single-ion conducting polymer electrolytes (SPEs) are provided. A first group of polymers are polysiloxanes substituted with fluorinated poly(alkylene oxide) side chains having associated ionic species, while a second group are copolymers containing mer units having structures (II) and (III), in which R4 through R8, x2, x3, y2, y3, z2 and z3 are as defined herein. Also provided are conductive compositions containing these novel polymers, particularly film compositions, and batteries formulated with such films.



$$\frac{1}{\left(0\right)_{12} - \left(CH_{2}\right)_{12} - \left(\frac{R^{6}}{R^{8}} - \left(CH_{2}\right)_{12} - \frac{1}{R^{6}} \right)}$$
 (II)

$$\frac{\begin{bmatrix} & & & & \\ & & & \\ & & & \\ & & & \end{bmatrix}}{\begin{bmatrix} & & & \\ & & \\ & & \\ & & & \end{bmatrix}} (III)$$

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SINGLE-ION CONDUCTING SOLID POLYMER ELECTROLYTES

Technical Field

The present invention relates generally to novel single-ion conducting polymer electrolytes. The invention additionally relates to conductive compositions containing these novel polymers in combination with one or more plasticizers. Also within the scope of the present invention are single-ion conducting gel electrolyte film compositions, and methods of manufacturing such compositions. The invention further relates to the use of these polymers and conductive compositions in solid-state batteries, fuel cells, sensors, supercapacitors, electrochromic devices and the like.

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Background of the Invention

A number of solvent-free polymer electrolytes are known and there has been considerable interest in the potential use of such electrolytes in electrochemical devices such as solid-state batteries, fuel cells, sensors, supercapacitors and electrochromic devices. Polymer electrolytes in general have a number of desirable features, i.e., they are inherently safe in operation, they avoid the leakage and drying problems experienced with liquid compositions, and they are further relatively processable. An additional advantage of solid polymer electrolytes is their ability to deform and thus maintain interfacial contact with electrodes. Finally, polymer electrolytes may be cast in thin films

to minimize resistance of the electrolyte and to reduce volume and weight.

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Among the polymers which have been tested for use in solvent-free electrolyte systems are those based upon the linear-chain polyethers, poly(ethylene oxide) ("PEO") and poly(propylene oxide) ("PPO"), with associated alkali metal salts such as lithium salts. Representative PEO and PPO polymers are described by Le Nest et al., in Polymer Communications 28:302-305 (1987) and by Tsuchida et al., Macromolecules 88:96-100 (1988). However, such electrolytes display conductivity in the range of practical use (e.g., $\sigma = 10^{-5}-10^{-3}$ S/cm) only at temperatures well above room temperature. Further, the reported linear-chain polyether electrolytes exhibit an ion transport number that is significantly lower than one, as both the anion and cation have ionic mobility and eventually account for the polymer electrolyte conductivity. Accordingly, a considerable amount of research has been focused on providing conductive solid polymer electrolytes capable of exhibiting conductivities in the range of their liquid electrolyte counterparts.

Attempts at improving the ionic conductivity of such polymer electrolytes have included the synthesis of new polymeric materials such as cation conductive phosphazene and siloxane polymers which exhibit better conductivity at room temperature than the linear-chain polyether electrolytes. In this regard, one class of polymers of interest are the polyphosphazene sulfonates as reported by Ganapathiappan et al. in both Macromolecules 21:2299-2301 (1988) and the Journal of the American Chem. Soc. 111:4091-4095 (1989); see also Chen et al., Chem. of Materials 1:483-484 (1984).

Other attempts at improving ionic conductivity have dealt with comb-like polymers with oligo-oxyethylene side chains anchored to a polyphospazene,

polymethacrylate or polysiloxane backbone. See, e.g., Blonsky et al., J. Am. Chem. Soc. 106:6854-6855 (1984), Bannister et al., Polymer 25:1600-1602 (1984) and Spindler et al., J. Am. Chem. Soc. 110:3036-3043 (1988). Since the movement of ions through the polymer matrix proceeds essentially by a free volume mechanism, polymers with flexible side chains are generally preferred. Cation transport polymer electrolytes based on cation conductive siloxane comb polymers are reported by Zhou et al., Poly. Comm. 30:52-55 (1989) and by Rietman et al., J. of Poly. Sci: Part C: Polymer Letters 28:187-191 (1990). Solid polymer electrolytes having anionic conductivity have been reported as well, see, e.g.,

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In solid electrolytic systems, single-ion conductive polymers provide a distinct advantage over dual-ion conductive polymers (wherein both the anion and cation have mobility in the electrolyte) in that they can charge and discharge more completely (in part because DC polarization does not occur). More particularly, single-ion conducting polymer electrolytes have the capability of exclusively transporting cations, such as lithium, thereby minimizing polarization effects at the electrodes. Further, single-ion conducting electrolytes avoid the condition wherein both the dissociated cation and anion of the metal salt dissolve in the electrolyte and move toward the positive and negative electrodes at the same time, reducing the ion transportation value.

Miyanishi et al., Macromolecules 17:975-977 (1984).

A number of single-ion conducting electrolytes

30 have been reported. Poly(ethylene oxide)—polyelectrolyte
blends--consisting of PEO mixed with an acrylate polymer
having pendant sulfonate or perfluorocarboxylate groups-have been described which exhibit a lithium ion
transference number close to unity. See, e.g., Bannister

35 et al., Polymer 25:1291-1296 (1984). A single-ion

conducting solid polymer electrolyte system comprising a solid solution having ionic species dissolved therein has also been described in U.S. Patent No. 5,102,751 to Narang et al., the disclosure of which is incorporated herein by reference. Further, a single-ion conducting polymer consisting of short PEO units functionalized by N-(fluoroalkylsulfonate) amido has been reported. See, e.g., Armand et al., (Seventh International Meeting on Lithium Batteries), May 15-20, 1994. However, each of the above-described single-ion conducting polymer systems generally exhibit low conductivity (e.g., $\sigma \leq 10^{-5}$ S/cm at 100°C) as well as low electrochemical stability.

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Accordingly, while the various solid polymer electrolytes set forth in the above publications have shown promise, those materials have limitations which prevent them from practical use in, for example, high energy-rechargeable batteries and other applications in which high ionic conductivity is necessary and wherein relatively thin films of the polymer electrolyte must be prepared. As noted above, prior polymer electrolytes do not exhibit sufficient ionic conductivity, particularly at room temperature. Further, such prior polymer electrolytes have generally not exhibited desirable physical properties for incorporation in electrolytic devices where, frequently, thin films of these electrolytes are necessary. For example, physical limitations inherent in those polymers include polymer films which may be too sticky, the polymers may be too close to being liquid, the polymers may be too brittle, or the polymers may be too heat sensitive.

One approach to overcoming some of the abovenoted problems (i.e., brittleness, low ionic conductivity, and the like) with prior polymer electrolytes has been the combination of those electrolytes with liquid electrolytes that serve as

plasticizers. In this manner, a number of plasticizers have been found to be useful in enhancing the ionic conductivity of solid polymer electrolytes. See, e.g., U.S. Patent 5,102,751 to Narang et al., incorporated by reference above. Additionally, gel electrolytes 5 containing poly(vinylidene fluoride) ("PVdF") have been developed, although such polymer electrolytes generally contain conventional lithium salts which are known to behave as dual-ion conductors, reducing the cation 10 transport values obtainable from such systems. gel electrolytes containing plasticizers have been reported (see, e.g., Tsuchida et al., Electrochemical Acta 28(5):591-595 (1983)); however, such electrolytes have been found to exhibit insufficiently high conductivity at room temperature. 15 The use of PVdF copolymers to prepare gel electrolytes containing lithium salts has also been described by Gozdz et al. in U.S. Patent No. 5,296,318; however, that method did not enable preparation of homogeneous, physically strong gel 20 electrolyte films without phasic separation of the lithium salt.

Accordingly, although some prior plasticizers have been shown to improve conductivity in solid electrolyte polymers, those compositions still suffer from serious drawbacks. In particular, prior plasticizers have been found to be too volatile, causing them to separate from the polymer electrolyte composition over a period of time. Such separation results in a decrease in the conductivity, and further, the physical properties of the polymer will likewise change; for example, the polymer might become more brittle and/or might peel from a substrate on which it has been coated.

Other prior liquid electrolytes and plasticizers, such as propylene carbonate, are known to be reduced at the lithium anode or carbon anode of

lithium batteries, therefore limiting battery performance. See, e.g., Arakawa et al., J. Electroanal. Chem. 219:273-280 (1987) and Shu et al., J. Electrochem. Soc. 140(4):922-927 (1993). The extent of propylene carbonate reduction is particularly severe on graphite electrodes. Although crown ethers have been used as additives in batteries to minimize such propylene carbonate reduction at the anode (see, e.g., Shu et al., J. Electrochem. Soc. 140(6):L101-L103 (1993) and U.S. Patent No. 5,130,211 to Wilkinson et al.), high 10 concentrations (0.3-0.5 M) of crown ether are needed to adequately minimize electrolyte reduction. In this regard, since crown ethers are highly toxic and generally quite expensive, they are not expected to be of practical use in batteries. 15

Accordingly, there remains a need to provide single-ion conductive solid polymer electrolytes capable of exhibiting conductivities in the range of their liquid electrolyte counterparts at room temperature (e.g., in the range of $\sigma \geq 10^{-3}$ S/cm at 20°C) as well as enhanced electrochemical stability. Additionally, there has remained a need to develop plasticizers for use with such polymer electrolytes that are capable of providing a plasticizing effect while also significantly enhancing the ionic conductivity of the solid polymer. Such plasticizers should not exhibit the drawbacks experienced by prior systems such as being readily volatilized away from the polymer and/or deleteriously altering the mechanical properties of the polymer.

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Summary of the Invention

In accordance with an embodiment of the present invention, solid polymer electrolytes are set forth having single-ion cationic conductivity and exhibiting enhanced ambient temperature ionic conductivity. The

single-ion conducting polymers as described herein include polysiloxanes, partially fluorinated polymethacrylates and poly(alkylene oxide) solid polymer electrolytes ("SPEs") which are covalently functionalized by fluoroalkylsulfonate groups.

In one embodiment of the invention, single-ion conducting polymers are provided having the structure

(1)
$$\frac{\begin{bmatrix} R^2 \\ si \\ R^1 \end{bmatrix} - o = \begin{bmatrix} R^2 \\ si \\ R^1 \end{bmatrix}$$

wherein:

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 \mathbb{R}^1 and \mathbb{R}^2 are individually selected from the group consisting of moieties having the structure

$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2R^3$$

in which R^3 is -OM, -N(M)SO₂CF₃ or -C(M)(SO₂CF₃)₂ and M is an alkali metal, or wherein one of R^1 and R^2 has the structure

$$-(CH2)x1(OCH2CH2)y1(OCF2CF2)z1SO2R3$$

and the other is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene; x1 and z1 may be the same or different and are integers in the range of 1 to 100 inclusive;

yl is an integer in the range of 0 to 100 inclusive; and

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n is an integer indicating the number of mer units in the polymer.

In another embodiment of the invention, singleion conducting co-polymers are provided which contain first mer units having the structure

(II)
$$\frac{1}{\left((CH_2)_{y2} - \frac{R^4}{C} - (CH_2)_{z2} - \frac{R^4}{C} \right)}$$

and second mer units having the structure

15 (III)
$$\frac{1}{(CH_2)_{y3}} - \frac{R^6}{C} - (CH_2)_{z3} = \frac{1}{R^7}$$

wherein:

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 R^4 and R^6 are independently selected from the group consisting of hydrogen, lower alkyl, lower alkoxy, $-\text{COOR}^8$ and $-(\text{CH}_2)_{n1}-\text{O-R}^8$ in which R^8 is lower alkyl or fluorinated lower alkyl and n1 is an integer in the range of 1 to 6 inclusive;

 $\rm R^5$ is $-(\rm CH_2)_{x4}(\rm OCH_2CH_2)_{y4}(\rm OCF_2CF_2)_{z4}SO_3M$ in which M is an alkali metal;

 $\rm R^7$ is $-(\rm CH_2)_{x5}(\rm OCH_2CH_2)_{y5}\rm OCH_3$ or $-\rm COOR^9$ in which $\rm R^9$ is lower alkyl or fluorinated lower alkyl; and

x2, x3, x4, x5, y2, y3, y4, y5, z2, z3 and z4 may be the same or different and are integers in the range of 1 to 100 inclusive.

In a further embodiment of the present invention, conductive compositions having enhanced ambient temperature conductivity are provided, where

those compositions are formed from a combination of a single-ion conducting SPE according to the present invention, and an amount of a plasticizer effective to enhance the ionic conductivity of the associated polymer. More particularly, conductive compositions containing the single-ion conducting polymers of the present invention as described above in combination with suitable medium-boiling solvents or liquid electrolytes are disclosed herein which exhibit high ambient temperature ionic conductivities and excellent physical and mechanical attributes such as high flexibility, strength and electrochemical stability. Suitable plasticizers generally feature a high dielectric constant, a medium to high boiling point, and a low viscosity.

Plasticizers which are specifically contemplated for use herein include medium-boiling solvents such as ethylene carbonate ("EC"), propylene carbonate ("PC") and combinations thereof such as a 1:1 mixture by weight of PC:EC. Other medium-boiling organic solvents which are contemplated for use as plasticizers herein include other lower alkyl carbonates such as dimethyl carbonate, diethyl carbonate and dipropyl carbonate, and glymes (e.g., dimethoxyethane ($C_4H_{10}O_2$ or "DME"), diglyme ($C_6H_{14}O_3$), triglyme ($C_8H_{18}O_4$), tetraglyme ($C_{10}H_{22}O_5$) and so on); either alone or in combination. Cyclic carbonates may also be used as plasticizers in combination with the present polymers, e.g., functionalized cyclic ethers having the general structure

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wherein:

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 R^{10} is alkyl, $-(OC_2R^{11}_4)_m-R^{12}$ or $-(CO)OR^{13}$; R^{11} is independently selected from the group consisting of H, alkyl, aryl, alkenyl, fluoro, or fluorinated alkyl;

 R^{12} is H, alkyl or $-(OC_2R^{11}_4)_m-R^{13}$; R^{13} is H or alkyl;

p is an integer in the range of 1 to 5 inclusive; and

q is an integer in the range of 0 to 6 inclusive.

Still other plasticizers contemplated for use herein are described in co-pending U.S. Patent Application Serial No. 08/372,193, entitled "ORGANIC LIQUID ELECTROLYTES AND PLASTICIZERS" inventors Ventura et al., filed on even date herewith and incorporated herein by reference.

In yet a further embodiment of the present invention, single-ion conducting gel electrolyte film compositions are provided wherein those films are generally formed from a combination of a single-ion conducting SPE as described above with a plasticizer and a suitable amount of poly(vinylidene fluoride) for enhancing the mechanical strength of the resulting gel electrolyte composition. In this manner, the single-ion conducting SPEs of the present invention exhibit desirable physico-mechanical properties in that they may be formulated in thin but nevertheless highly conducting films having desirable physical properties such as enhanced mechanical strength and lack of stickiness.

According to the invention, there is further described a method of manufacturing single-ion conducting electrolyte films and other conductive compositions.

Generally, such a method will involve a hot press technique for forming films; however, depending on the

amounts of various components incorporated into the compositions, waxes and gels may be prepared as well.

In still another embodiment of the present invention, the SPEs may be used in the formulation of solid-state electrochemical devices such as fuel cells, supercapacitors, electrochromic devices and sensors, or in a battery such as in a solid state lithium battery or the like. In this regard, a solid-state battery comprising a positive electrode, a negative electrode and a single-ion conducting solid polymer electrolyte as described above is disclosed herein.

In yet a further embodiment of the invention, the presently described SPEs may be used in combination with plasticizers to form thin film SPEs; and, optionally, PVdF or an alternative material may be added in order to improve the SPE film mechanical strength. Accordingly, also provided herein are solid-state batteries comprising a positive electrode, a negative electrode and a single-ion conducting polymer electrolyte film according to the invention. With the subject SPE films, solid-state batteries such as a Li_xC₆/SPE/LiCoO₂ battery may be fabricated having enhanced conductivity and rechargability.

25 Brief Description of the Figure

FIG. 1 is a diagram illustrating ranges of components in a composition containing a single-ion conducting polymer electrolyte, a plasticizer, and PVdF, for forming films, waxes and gels.

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<u>Detailed Description of the Invention</u> Definitions:

Before describing the present invention in detail, it is to be understood that this invention is not limited to particular salts, methods of synthesis, solvents, or the like, as such may vary. It is also to be understood that the terminology used herein is for the purpose of describing particular embodiments only, and is not intended to be limiting.

It must be noted that, as used in this specification and the appended claims, the singular forms "a," "an" and "the" include plural referents unless the content clearly dictates otherwise. Thus, for example, reference to "a plasticizer" includes mixtures of plasticizers, and the like.

In this specification and in the claims which follow, reference will be made to a number of terms which shall be defined to have the following meanings:

The term "polymer" is intended to include both oligomeric and polymeric species, i.e., compounds which include two or more monomeric units, which may be a homopolymer or a copolymer. When a single generic structure is shown, e.g., as in formula (I), it is to be understood that the polymers described may contain two or more different monomeric units represented by the single generic structure. A "conductive polymer" is one which possesses conducting as opposed to insulating electrical-transport properties.

The term "homopolymer" intends a polymer

incorporating a single species of monomer units. By
contrast, the term "copolymer" refers to a polymer
constructed from two or more chemically distinct species
of monomer units in the same polymer chain. A "block
copolymer" is a polymer which incorporates two or more

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segments of two or more distinct species of homopolymers or copolymers.

The term "alkyl" as used herein refers to a branched or unbranched saturated hydrocarbon group of 1 to 24 carbon atoms, such as methyl, ethyl, n-propyl, isopropyl, n-butyl, isobutyl, t-butyl, octyl, decyl, tetradecyl, hexadecyl, eicosyl, tetracosyl and the like. Preferred alkyl groups herein contain 1 to 12 carbon atoms. The term "lower alkyl" intends an alkyl group of one to six carbon atoms. The term "fluorinated lower alkyl" intends an alkyl group of one to six carbon atoms in which at least one hydrogen atom, and optionally all hydrogen atoms, are replaced with fluorine atoms.

The term "alkenyl" refers to a branched or unbranched hydrocarbon chain containing from 2 to 24 carbon atoms and at least one double bond. "Lower alkenyl" refers to an alkenyl group of 2 to 6, more preferably 2 to 5, carbon atoms. The term "fluorinated lower alkenyl" intends an alkenyl group of one to six carbon atoms in which at least one hydrogen atom, and optionally all hydrogen atoms, are replaced with fluorine atoms.

The term "alkoxy" as used herein intends an alkyl group bound through a single, terminal ether linkage; that is, an "alkoxy" group may be defined as -OR where R is alkyl as defined above. A "lower alkoxy" group intends an alkoxy group containing one to six, more preferably one to four, carbon atoms.

The term "aryl" as used herein refers to a monocyclic aromatic species of 5 to 7 carbon atoms, and is typically phenyl. Optionally, these groups are substituted with one to four, more preferably one to two, lower alkyl, lower alkoxy, hydroxy, and/or nitro substituents or the like.

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The term "aralkylene" is used to refer to moieties containing both alkylene and monocyclic aryl species, typically containing less than about 12 carbon atoms in the alkylene portion, and wherein the aryl substituent is bound to the structure of interest through an alkylene linking group. Exemplary aralkylene groups have the structure -(CH₂)_j-Ar wherein j is an integer in the range of 1 to 6.

"Halo" or "halogen" refers to fluoro, chloro, bromo or iodo, and usually relates to halo substitution for a hydrogen atom in an organic compound. Of the halos, fluoro is typically preferred.

"Optional" or "optionally" means that the subsequently described circumstance may or may not occur, and that the description includes instances where said circumstance occurs and instances where it does not. For example, the phrase "optional covalent bond" means that a covalent bond may or may not be present and that the description includes both the instance when the covalent bond is present and the instance when the covalent bond is not present.

The Novel Compounds:

In one embodiment of the invention, there are provided a number of polysiloxane single-ion conducting solid polymer electrolytes covalently functionalized by fluoroalkylsulfonate groups. More particularly, there are provided polysiloxane SPEs having the general structure as shown in Formula I

$$(1) \qquad \qquad \frac{\begin{bmatrix} R^2 \\ 1 \\ R^1 \end{bmatrix} - 0 = \begin{bmatrix} R^2 \\ R \end{bmatrix}}{n}$$

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wherein:

 ${\ensuremath{\mathsf{R}}}^1$ and ${\ensuremath{\mathsf{R}}}^2$ are individually selected from the group consisting of moieties having the structure

-(CH₂)_{x1}(OCH₂CH₂)_{y1}(OCF₂CF₂)_{x1}SO₂R³

in which R^3 is -OM, $-N(M)SO_2CF_3$ or $-C(M)(SO_2CF_3)_2$ and M is an alkali metal, or wherein one of R^1 and R^2 has the structure

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$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2R^3$$

and the other is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene; x1 and z1 may be the same or different and are integers in the range of 1 to 100 inclusive;

y1 is an integer in the range of 0 to 100 inclusive; and

n is an integer indicating the number of mer units in the polymer, generally although not necessarily providing the polymer with a (weight average) molecular weight in the range of about 10,000 to 3,000,000, more typically in the range of about 100,000 to 1,000,000.

In preferred polysiloxane SPEs having the structure shown in Formula I, M is lithium.

In one group of particularly preferred polysiloxane SPEs having the structure of Formula I, \mathbb{R}^1 and \mathbb{R}^2 are the same moiety having the structure

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$$-(\mathtt{CH_2})_{\,\mathtt{x}\,\mathtt{1}}\,(\mathtt{OCH_2CH_2})_{\,\mathtt{y}\,\mathtt{1}}\,(\mathtt{OCF_2CF_2})_{\,\mathtt{z}\,\mathtt{1}}\mathtt{SO_3Li}$$

in which x1 and z1 may be the same or different and are integers in the range of 1 to 100 inclusive, and y1 is an integer in the range of 0 to 100 inclusive;

In another group of particularly preferred SPEs encompassed by Formula I, \mathbb{R}^1 is a moiety having the structure

-(CH₂)_{x1}(OCH₂CH₂)_{y1}(OCF₂CF₂)_{z1}SO₃Li

in which x1, y1 and z1 are as defined above, and R^2 is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene.

In still another group of particularly preferred SPEs having the structure of Formula I, \mathbb{R}^1 and \mathbb{R}^2 are the same moiety having the structure

15 $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2N(Li)SO_2CF_3$

in which x1, y1 and z1 are as defined above.

Additional particularly preferred SPEs of Formula I are wherein \mathbb{R}^1 is a moiety having the structure

 $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2N(Li)SO_2CF_3$

R² is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene, and x1, y1 and z1 are as defined above.

Still other particularly preferred SPEs having the structure of Formula I are wherein \mathbb{R}^1 and \mathbb{R}^2 are the same moiety having the structure

 $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2C(Li)(SO_2CF_3)_2$

in which x1, y1 and z1, again, are as defined above.

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Yet an additional group of particularly preferred SPEs according to Formula I are those wherein \mathbb{R}^1 is a moiety having the structure

$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2C(Li)(SO_2CF_3)_2,$$

R² is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene, and x1, y1 and z1 are as defined earlier.

In additional to the polysiloxane family of SPEs provided herein, i.e., those polymers having the structure of Formula I, a number of polymethacrylates and poly(alkene oxides) covalently functionalized by fluoroalkylsulfonate groups have been found to be useful as well. More particularly, there are provided a number of copolymeric SPEs containing first mer units having the structure (II)

(II)
$$\frac{\mathbb{R}^4}{\mathbb{C}^{(CH_2)_{y2}} - \mathbb{C}^{(CH_2)_{z2}}}$$

and second mer units having the structure (III)

(III)
$$\frac{1}{(0)_{x3} - (CH_2)_{y3}} - \frac{R^6}{C} - (CH_2)_{z3}$$

wherein:

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 R^4 and R^6 are independently selected from the group consisting of hydrogen, lower alkyl, lower alkoxy, $-\text{COOR}^8$ and $-(\text{CH}_2)_{n1}$ -O- R^8 in which R^8 is lower alkyl or

fluorinated lower alkyl and n1 is an integer in the range of 1 to 6 inclusive;

 $\rm R^5$ is $-(\rm CH_2)_{\,x4}\,(\rm OCH_2CH_2)_{\,y4}\,(\rm OCF_2CF_2)_{\,z4}SO_3M$ in which M is an alkali metal;

 $\rm R^7$ is $-(\rm CH_2)_{x5}(\rm OCH_2CH_2)_{y5}\rm OCH_3$ or $-\rm COOR^9$ in which $\rm R^9$ is lower alkyl or fluorinated lower alkyl; and

x2, x3, x4, x5, y2, y3, y4, y5, z2, z3 and z4 may be the same or different and are integers in the range of 1 to 100 inclusive.

As with the polymers of Formula I, preferred copolymers containing mer units (II) and (III) are wherein M is lithium. Also, these copolymers will generally have a (weight average) molecular weight in the range of about 10,000 to 3,000,000, more typically in the range of about 100,000 to 1,000,000.

A number of preferred copolymers useful as SPEs may be identified. A first group of such copolymers are wherein R^4 and R^6 are hydrogen or lower alkyl, and R^7 is $-(CH_2)_{x4}(OCH_2CH_2)_{y4}OCH_3$, with x4 and y4 defined as above.

Another group of particularly preferred copolymers useful as SPEs are wherein ${\bf R}^4$ and ${\bf R}^7$ are independently -COOR⁸ and ${\bf R}^6$ is hydrogen or lower alkyl.

The polymers of the invention may be prepared using conventional techniques well-known to those skilled in the art of synthetic organic chemistry or which may be found in the relevant texts such as in Kirk-Othmer's Encyclopedia of Chemical Technology, in House's Modern Synthetic Reactions, in C.S. Marvel and G. S. Hiers' text, ORGANIC SYNTHESIS, Collective Volume 1, or the like. Synthesis of representative polymers is exemplified below.

Manufacturing Methods:

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A preferred method of manufacturing conductive compositions containing the novel electrolytes is a hot-

press technique for forming films. Such a method typically involves: (a) forming a gel electrolyte composition by combining (i) a single-ion conducting SPE of the invention (i.e., a polysiloxane of Formula (I) or a copolymer containing mer units (II) and (III)), with (ii) an effective amount of plasticizer for enhancing the ionic conductivity of that SPE and (iii) an amount of PVdF or an alternative material effective to enhance the mechanical strength of the composition; (b) heating the resulting combination at a temperature and for a time effective to form a fluid solution; (c) pressing the fluid solution; (d) cooling the solution; and (e) releasing the film so provided.

If waxes or gels are preferred, rather than films, the relative quantities of components can be adjusted to provide these alternative forms of conductive compositions. Reference may be had to Figure 1, in which it may be seen that compositions containing less electrolyte will generally form a gel, compositions containing slightly more electrolyte will generally form a wax, and compositions containing even more electrolyte will form a film. Alternative methods of manufacturing such conductive compositions will be readily apparent to those skilled in the art, or may be deduced from the relevant literature.

Industrial Applicability:

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Conductive compositions formulated with the novel single-ion conducting solid polymer electrolytes are useful in a variety of contexts. An important utility is in the fabrication of batteries. Solid-state batteries formulated with the novel electrolyte polymers comprise a positive electrode, or anode, a negative electrode, or cathode, and the single-ion conducting polymer electrolyte. The anode is usually a carbon-based

material such as petroleum coke or graphite. Alternatively, lithium metal may be used as the anode, or intercalating metal oxides such as tungsten or iron oxides. The cathode is generally of a lithium-containing material such as LiCoO_2 , LiMn_2O_4 or LiNiO_2 ; however, alternative materials could be used as well, e.g., V_6O_{13} .

It will be appreciated that conductive compositions formulated with the novel electrolyte materials of the invention are also useful in the fabrication of fuel cells, sensors, supercapacitors, electrochromic devices, and the like, using manufacturing techniques well known to those skilled in the art, or readily available in the relevant literature.

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The following examples are intended to provide those of ordinary skill in the art with a complete disclosure and description of how to make and use the materials of the invention, and are not intended to limit the scope of what the inventors regard as their invention in any way. Efforts have been made to ensure accuracy with respect to numbers used (e.g., amounts, temperatures, etc.), but some experimental error and deviation should, of course, be allowed for. Unless indicated otherwise, parts are parts by weight, temperatures are in degrees centigrade, and pressure is at or near atmospheric.

Experimental

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Equipment and Measurement Techniques:

conductivities of the polymers were evaluated by AC impedance spectroscopy. In this regard, a film of the dried polymer electrolyte was sandwiched between two stainless steel blocking electrodes, each having an area of 0.7854 cm². The thickness of the polymer film, which typically varied between 0.51 mm and 1.02 mm, was measured with a micrometer. An assembly, composed of a blocking electrode-polymer sandwich cell inside a Delrin cup, was transferred to a vacuum chamber that had provision for heating and applying a constant pressure of 65-97 lb/in₂ across the polymer film. Electrodes were connected to a potentiostat (PAR 173) operating in the galvanostatic mode.

The cell was then perturbed with a small AC signal generated by a Solartron 1250 Frequency Response Analyzer, and the real and imaginary components of the cell impedance were measured as a function of frequency at each of the desired temperatures. The setup was allowed to stabilize overnight after the temperature was changed. The AC impedance data were plotted in both the Nyquist and Bode planes to identify the high frequency relaxation arising due to the polymer electrolyte. Typically, the frequency of the AC signal was scanned from 65 KHz down to 10 mHz. The intercept at the real axis of the high frequency relaxation was assumed to represent the resistive component of the polymer electrolyte impedance. This was then converted to the resistivity of the polymer (the thickness and the area of the polymer films were known). The reciprocal of resistivity gave the conductivity, σ , having units of Ω -cm⁻¹. In cases where high frequency relaxation occurred above 65 KHz, a Hewlett-Packard 4192A Impedance

Analyzer was used to measure the polymer electrolyte resistance. This instrument has a frequency range capability of 13 MHz to 5 Hz.

5 Preparation of Polymer Films:

Solutions of polymer films were prepared by dissolving a known quantity of polymer in dry solvent. For conductivity measurements, the polymer solution was added dropwise into the Delrin cup to cast a film. The film was then dried for 3 days in a glass vacuum apparatus at 120°C at <0.01 torr. Film thickness was measured using a micrometer.

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Example 1 Preparation of Polymethylsiloxane with C3H6OC2F4SO3Li Pendant Group

Polymethylsiloxane carrying a $C_3H_6OC_2F_4SO_3Li$ pendant group was synthesized as according to the following scheme and procedures:

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$$CH_2 = CHCH_2Br + CF_2 - O - CH_2 = OHCH_2OC_2F_4SO_2F$$

 $CF_2 - SO_2$ (3)

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$$(6) + (CH3)3SiOSi(CH3)3 ----> (CH3)3SiO-(SiO)n-Si(CH3)3
C3H6OC2F4SO2F$$
(7)

(7) + 2 LiOH ----->
$$CH_3$$
 | CH_3 |

(a.) Preparation of Fluorosultone (2):

In a 500-ml Parr pressure reactor equipped with a magnetic stirrer, 50 ml of fresh sulfur trioxide was prepared by distillation of fuming sulfuric acid. Under stirring, tetrafluoroethylene was introduced continuously at the pressure of 30 psi. Exothermic reaction took

35 place. The volume of the liquid content increased

gradually as the reaction proceeded. At the end of the reaction, the crude product was purified by distillation. A colorless liquid product was collected at the temperature of 42-43°C. 162.9 g fluorosultone (2) was obtained.

(b.) Preparation of (3):

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In a 250-ml 3-neck flask equipped with a magnetic stirrer and covered with aluminum foil, 63.44 g (0.50 mol) of silver fluoride and 100 ml of anhydrous diglyme were combined. The flask was cooled to -78°C and 90.04 g (0.50 mol) of fluorosultone (2) was added dropwise. A clear solution was produced after 1 hr at room temperature. The flask was again cooled to -78°C and 60.49 g (0.50 mol) of allyl bromide (1) was added The reaction was heated at 45-50°C for 16 hours. The mixture was then filtered to remove AgBr. The filtrate was poured into 100 ml water and the oily layer which formed was washed three times with water and dried over MgSo₄. Distillation gave 79.16 g (3), 76.1%, bp 120-121°C. The identity of the product was confirmed using ¹H NMR spectroscopy (¹H NMR (CDCl/6): 4.60 (d, 2H, $CH_2=CH_2CH_2O-)$; 5.34-5.47 (m, 2H, $\underline{CH_2}=CH-)$; 5.89-5.98 (m, 1H, $CH_2 = CHCH_2O-)$).

(c.) Preparation of (5):

In a high pressure reactor, 72.69 g (0.35 mol) of (3), 80.53 g (0.70 mol) dichloromethylsilane (4) and 0.46 g (1.1 mol) catalyst chloroplatinic acid were combined. The reactor was sealed and filled with argon to a pressure of 50 psi and then heated at 70-90°C for 22 hrs. After cooling to room temperature, the product was then transferred under inert atmosphere to a flask using a double-tip needle. Un-reacted dichloromethylsilane was removed under reduced pressure. Distillation gave 85.89 g (5), 76%, bp 68-71°C/0.80 mm Hg. The identity of the product was confirmed using ¹H NMR spectroscopy (¹H NMR)

 $(CDCl_3/\delta): 0.82 (s, 3H, CH_3Si-); 1.15-1.25 (m, 2H,$ SiCH2CH2CH2-); 1.9-2.0 (m, 2H, SiCH2CH2-); 4.1-4.2 (t, 2H, $sich_2ch_2ch_2oc_2F_4so_2F$)).

(d.) Preparation of (6):

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(5) (85.8 g, 0.27 mol) was taken up in 5 . anhydrous ether (50 mL), and the resultant solution was then added to a mixture of water (50 mL) and ether (50 mL) dropwise under stirring. At the end of the addition, the reaction was allowed to continue overnight. layer was separated from the aqueous layer. The aqueous layer was then extracted with ether (40 mL x 3), and the ether extract was combined with the oily layer product, washed with water until the water phase became neutral to pH paper, dried over anhydrous MgSo4, filtered, and the solvent was removed from the filtrate by rotary evaporator. The resulting colorless liquid residue was further dried at room temperature under 0.1 torr vacuum for 4 hours to yield 70.68 g. cyclosiloxane (6) (99%). The identity of the intermediate (6) was confirmed using 1 H NMR spectroscopy (1 H NMR (CDCl $_{3}/\delta$): 0.05-0.08 (m, 3H, CH₃Si-); 0.25-0.35 (m, 2H, SiCH₂CH₂CH₂-); 1.85-1.90 (broad, 2H, SiCH₂CH₂O-); 4.0-4.5 (broad, 2H, $SiCH_2CH_2CH_2O)$).

(e.) Preparation of (7):

The cyclosiloxane (6) (350 g, 1.305 mol) and hexamethyldisiloxane (42.4 g, 0.261 mol) were added to a round bottom flask with 10 drops of concentrated sulfuric acid and allowed to sit overnight. An additional 10 drops of concentrated sulfuric acid were then added and the mixture was stirred for 24 hours. The mixture was then taken in methylene chloride and washed with water (2 x 500 mL) and then a saturated solution of NaCl containing a small amount of NaHCO3. dried over MgSO, before removing the solvent by evaporation.

(f.) Preparation of (8):

The sulfonyl fluoride polysiloxane (7) (59.80 g, 0.22 mol) was taken up in 270 mL THF. To the THF solution, lithium hydroxide aqueous solution (10.67 g of anhydrous LiOH, 0.44 mol, in 150 mL water) was added dropwise in 6 hours. Theoretically, two moles of lithium hydroxide are required to convert each sulfonyl fluoride group to lithium sulfonate. However, lithium hydroxide readily absorbs moisture because it is hygroscopic. A slight excess of LiOH was added until the solution became 10 neutral using a pH meter as a monitor. The absence of ¹⁹F NMR absorption at 123.5 ppm due to -SO₂F confirmed that all of the sulfonyl fluoride groups were converted to lithium sulfonate. THF and water were then rotary evaporated off. The residual white solid was dried at 15 50°C under 0.1 torr vacuum overnight.

The resulting crude product was dissolved in 200 mL acetone. The mixture was filtered in order to remove LiF, and the filtrate was then concentrated. The concentrated acetone solution was added dropwise to 600 mL of dry ether under stirring. The polymer product was precipitated out. The polymer product (8) was then dried at 70-80°C under 0.05 torr vacuum for 2 days, and the pure product (8) was obtained in 93% yield. The identity of the product (8) was confirmed using ¹H NMR spectroscopy (¹H NMR (D₂O/δ): 0.21 (broad, 3H, CH₃Si-); 0.70 (broad, 2H, SiCH₂CH₂CH₂-); 1.80 (broad, 2H, SiCH₂CH₂CH₂-); 4.10 (broad, 2H, SiCH₂CH₂CH₂COC₂F₄SO₃Li); ¹⁹F NMR (D₂O/ppm): -5.61 (s, -CF₂CF₂SO₃Li); -39.03 (s, -CF₂CF₂SO₃Li)).

Example 2

A lithium triflate polysiloxane single-ion conducting polymer having the general structure

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was synthesized using the method of the preceding example. The polymer was combined with various plasticizers, and the conductivity σ ($\Omega^{-1} cm^{-1}$) was determined using the method described above. The experimental results thus obtained are set forth in Table I.

25 30	qe 1 20	Table I 12	5
Plasticizer	Plasticizer Ratio	Ratio of Polymer: to Plasticizer	σ (n^{-1} cm ⁻¹)
12-Crown-4	6 9 8	1:3	8.87 E ⁻⁵
PC: EC	1:1	1:3	2.99 E ⁻⁴
PC: DME	1:1	1:3	4.80 E-4
EC: DME	1:1	1:3	7.51 E ⁻⁴
EC: DME	2:1	1:3	7.45 E-4
EC:MEE	1:2	1:3	5.91 E-4
EC:MEE	1:1	1:3	1.06 E ⁻³
EC: MEE	1:1	1:6	1.02 E ⁻³
PC: DEC	1:1	1:3	1.06 E-4
PC: DEC	1:1	1:2	9.33 E-5
PC: EC: 12-Crown-4	1:1:1	1:3	3.91 E-4
PC: EC: DEC	1:1:2	1:3	3.04 E-4
DEC: EC: MEE	1:1:2	1:3	6.94 E-4
PC: EC: DME	1:1:2	1:2	6.34 E-4
PC: EC: DME	1:1:2	1:2.5	8.71 E-4
PC: EC: DME	1:1:2	1:3	1.07 E ⁻³
PC: EC: MEE	1:1:2	1:3	1.11 E ⁻³
PC: EC: MEE	1:1:2	1:3.5	5.24 E-4
PC: EC: MEE	1:1:2	1:4	4.32 E-4

In Table I, the abbreviations are as follows:

PC, propylene carbonate; EC, ethylene carbonate; DME,
dimethoxyethane; MEE, methoxyethoxy ethyl ether; and
diethylene carbonate. It may be seen that high ambient
temperature conductivity may be achieved and varied, if
desired, by choosing different plasticizers and/or
relative quantities of plasticizer and polymeric
electrolyte in the conductive composition.

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Example 3

The polymers set forth in Table II were prepared by copolymerizing the corresponding monomeric species, and then combined with the plasticizers propylene carbonate, ethylene carbonate, and methoxyethoxy ethane and other materials as indicated in Table III. Conductivity data was determined and is set forth in Table III.

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Table II

	NUMBER	SAMPLE
5	SPE1	(-CH ₂ -CH-O-) _n CH ₂ OC ₂ F ₄ SO ₃ Li
10	SPE2	CH ₃ (-CH ₂ -C-CH ₂ -O-) _n CH ₂ OC ₂ F ₄ SO ₃ Li
15	SPE3	$(-CH_{2}-CH-O-)_{0.2}-(-CH_{2}-CH-O-)_{0.8}$ $CH_{2}OC_{2}H_{4}OC_{2}H_{4}OMe$ $CH_{2}OC_{2}F_{4}SO_{3}Li$
20	SPE4	$(-CH_{2}-CH-O-)_{0.2}$
	SPE5	$(-CH_{2}-CH-O-)_{0.5}$ $(-CH_{2}-CH-O-)_{0.5}$ $ $ $ $ $CH_{2}O(C_{2}H_{4}O)_{7.3}Me$ $CH_{2}SO_{3}Li$
25	SPE6	$(-CH_{2}-CH-O-)_{0.5}$ $(-CH_{2}-CH-O-)_{0.5}$ $ $ $CH_{2}O(C_{2}H_{4}O)_{7.3}Me$ $CH_{2}OC_{2}F_{4}SO_{3}Li$
	SPE7	CH ₂ OCF ₂ CF ₂ SO ₃ Li CH ₃
30		$(-CH_2-C)_x$ $(-CH_2-C)_y$ $CO_2CH_2CH_3$ $CO_2(CH_2)_3CH_3$

Table II (cont'd.)

5	NUMBER		SAMPLE	
10	SPE8		CH ₂ OCF ₂ CF ₂ SO ₃ Li -C-) _x (-CH ₂ CO ₂ CH ₂ CH ₃	H C-) _y Co ₂ CH ₃
15	SPE9	(-CH ₂ -	CH_3 $ $ $-C-)_x$ $-(-CH_2 $ $CO_2CH_2CH_2OCF_2CF_2SO_3Li$	CH ₂ OCH ₂ CF ₃ C-) _y CO ₂ CH ₂ CH ₃
20	SPE10	(-CH ₂ -	CH ₃ -C-) _x (-CH ₂ - CO ₂ CH ₂ CH ₂ OCF ₂ CF ₂ SO ₃ Li	CH ₂ OCH ₂ CF ₂ CF ₂ CF ₃ C-) _y co ₂ CH ₂ CH ₃
25	(-CH ₂ -	СН ₃ -С-) _ж СО ₂ СН	CH ₂ OCH ₂ 	$_{2}^{\text{CF}_{3}}$ $_{ }^{\text{CH}_{2}\text{OCH}_{2}\text{CF}_{2}\text{CF}_{2}\text{CF}_{3}}$ $ -(-\text{CH}_{2}- \text{C-})_{z}$ $ -(\text{CH}_{3} \text{CO}_{2}\text{CH}_{2}\text{CH}_{3}$
30	SPE12	(-СН ₂ -	$\begin{array}{c} \text{CH}_3 \\ \mid \\ -\text{C-})_x \end{array}$ $\left(-\text{CH}_2 - \text{CO}_2\text{CH}_2\text{CH}_2\text{OCF}_2\text{CF}_2\text{SO}_3\text{Li}} \right)$	H C-) _y CH ₂ OCH ₂ CH ₂ OCF ₂ CF ₂ SO ₃ Li
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Table II (cont'd.)

NUMBER	SAMPLE
SPE13	$(-CH_{2}-C-)_{x} \xrightarrow{CH_{2}-(-CH_{2}-C-)_{y}} (-CH_{2}-C-)_{y} \\ CO_{2}CH_{2}CH_{2}OCF_{2}CF_{2}SO_{3}Li CO_{2}CH_{2}CF_{3}$
SPE14	$(-CH_{2}-C-)_{x}$ $(-CH_{2}-C-)_{x}$ $(-CH_{2}-C-)_{y}$ $CO_{2}CH_{2}CH_{2}CCF_{2}CF_{2}SO_{3}Li$ $CO_{2}CH_{2}(CF_{2})_{6}CF_{3}$
SPE15	$(-CH_{2}-C-)_{x} \xrightarrow{CH_{2}OCH_{2}CF_{2}SO_{3}Li} CH_{2}OCH_{2}CF_{3}$ $(-CH_{2}-C-)_{x} \xrightarrow{(-CH_{2}-C-)_{y}} CO_{2}CH_{2}CH_{3}$
SPE16	$\begin{array}{cccccccccccccccccccccccccccccccccccc$
SPE17	$\begin{array}{cccccccccccccccccccccccccccccccccccc$

Table II (cont'd.)

NUMBER	SAMPLE
SPE18	$\begin{array}{cccccccccccccccccccccccccccccccccccc$
SPE19	$\begin{array}{c ccccccccccccccccccccccccccccccccccc$
SPE20	$\begin{array}{cccccccccccccccccccccccccccccccccccc$
SPE21	$\begin{array}{cccccccccccccccccccccccccccccccccccc$

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	_							-			T		T		T	T	T	
5	٠	σ (n ⁻¹ cM ⁻¹)	8.39 x 10 ⁻⁵	1.62 x 10 ⁻⁴	4.47 × 10 ⁻⁴	8.08 x 10 ⁻⁴	7.57 x 10 ⁻⁴	3.91 x 10 ⁻⁴	4.49 x 10 ⁻⁴	5.92 x 10 ⁻⁴	7.36 x 10 ⁻⁴	7.16 x 10 ⁻⁴	6.62 x 10 ⁻⁴	9.80 x 10 ⁻⁵ (~50% soluble)	4.22 x 10 ⁻⁴	insoluble	insoluble	2.17 × 10 ⁻⁴
10	III	RATIO OF POLYMER: PLASTICIZER	1:2	1:3	1:3	1:4	1:4.5	1:2	1:3	1:3.5	1:4	1:4.5	1:5	1:3	1:3	1:3	1:3	1:3
15	rable III	PLASTICIZER RATIO	1:1	1:1	1:1	1:1	1:1	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2	1:1:2
20		PLASTICIZER	PC: EC	PC: EC	EC:MEE	EC:MEE	EC: MEE	PC: EC: MEE	PC: EC: MEE	PC: EC: MEE	PC: EC: MEE	PC: EC: MEE	PC: EC: MEE	PC: EC: MEE				
25		A COMMAN	SPE1	SPF1	2051	CDF1	SPE1	SPE2	SPE3	SPE4	SPE5	SPE6						
30		S. C.	EAF NO.	2		4	ហ	9	, ,	. 0			2 -	12	13	14	15	16

5		Dioxane (Wt%)	49	53	50	45 36	43	41	34	36	49	49	44	0	46	52
		AIBN (mole%)	0.95	96.0	1.76	1.52	1.41	08.0	1.04	1.00	1.67	2.07	1.79	0.53	1.89	1.39
10	III (cont'd.)	Li Density (Wt%)	0.48	0.51	0.33	0.42	0.22	0.59	0.36	0.23	0.33	0.28	0.42		0.48	0.59
15	Table III (c	σ (n ⁻¹ CM ⁻¹)	4.14 x 10 ⁻⁴	5.87 x 10 ⁻⁴	4.55 E-4	8.54 E-4 4.99 x 10 ⁻⁴	2.20 E-4	4.85 E-4			5.86 E-4 3.62 E-4	2.68 E-4 4.21 E-4	2.20 E-4 5.80 E-4 (w/LiF)	4.67 E-4	5.12 E-4	5.57 E-4
20		Copoly- mer PC/EC/MEE	1:3	1:3	1:3	1:3	1:3	1:3	1:3		1:3	1:3	1:3	1:3	1:3	1:3
25		Copoly- mer Ratio	75:25	75:25	50:50	75:25 85:15	35:30:35	50:50	50:50	50:50	50:50	50:50	75:25	75:25 (H.MW)	85:15	50:50
30		Copoly- mer	SPE 7	SPE 8	SPE 9	SPE 10	SPE 11	SPE 12	SPE 13	SPE 14	SPE 15	SPE 16				SPE 17

5	
10	· :
15	Table III (cont'd.)
20	Tab
25	
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-yloc	Copoly- Copoly-		Li	20	30,0
mer	3	$\sigma (n^{-1} \text{ cm}^{-1})$	(Wt%)	(mole%)	(Wt%)
50:50	1.3	5.01 E-4	0.32	1.54	46
50.50	1.3	4.89 E-4	0.36	1.23	45
00.00		2.56 E-4	0.31	1.00	34
20.00			0.23	1.11	32

Claims:

1. A polymer having the structure

5 (I)
$$\frac{\begin{bmatrix} R^2 \\ Si \\ R^1 \end{bmatrix} = 0$$

wherein:

 R^1 and R^2 are individually selected from the group consisting of moieties having the structure

$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2R^3$$

in which R^3 is -OM, $-N(M)SO_2CF_3$ or $-C(M)(SO_2CF_3)_2$ and M is an alkali metal, or wherein one of R^1 and R^2 has the structure

$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2R^3$$

and the other is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl, and aralkylene;

x1 and z1 may be the same or different and are integers in the range of 1 to 100 inclusive;

y1 is an integer in the range of 0 to 100 inclusive; and

 \boldsymbol{n} is an integer indicating the number of mer units in the polymer.

- 2. The polymer of claim 1 wherein R^3 is -OM.
- 3. The polymer of claim 1 wherein \mathbb{R}^3 is $-N(M)SO_2CF_3$.

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4.	The	polymer	of	claim	1	wherein R	2	is
$-C(M)(SO_2CF_3)_2$.								

- 5. The polymer of claim 1 wherein M is 1 lithium.
 - 6. The polymer of claim 2 wherein M is lithium.
- 7. The polymer of claim 3 wherein M is lithium.
 - 8. The polymer of claim 4 wherein M is lithium.
- 9. The polymer of claim 2 wherein n is selected to provide the polymer with a weight average molecular weight \overline{M}_{W} in the range of approximately 10,000 to 3,000,000.
- 10. The polymer of claim 9 wherein n is selected to provide the polymer with a weight average molecular weight $\overline{\rm M}_{\rm w}$ in the range of approximately 100,000 to 1,000,000.
- - $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_3Li.$
 - 12. The polymer of claim 1 wherein \mathbb{R}^1 is a moiety having the structure
 - $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_3Li$

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and R^2 is a moiety selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene.

- 5 13. The polymer of claim 1 wherein \mathbb{R}^1 and \mathbb{R}^2 are the same moiety having the structure
 - $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2N(Li)SO_2CF_3.$
- 10 14. The polymer of claim 1 wherein \mathbb{R}^1 is a moiety having the structure
 - $-(CH_2)_{x1}(OCH_2CH_2)_{v1}(OCF_2CF_2)_{z1}SO_2N(Li)SO_2CF_3$
- and R² is a moiety selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene.
- 15. The polymer of claim 1 wherein \mathbb{R}^1 and \mathbb{R}^2 are the same moiety having the structure
 - $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2C(Li)(SO_2CF_3)_2.$
- 16. The polymer of claim 1 wherein \mathbb{R}^1 is a 25 moiety having the structure
 - $-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2C(Li)(SO_2CF_3)_2$
- and R² is a moiety selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene.
 - 17. A copolymer comprising first mer units having the structure

(II)
$$\frac{\left[(CH_2)_{y2} - \frac{R^4}{C} - (CH_2)_{z2} - \right]}{\left[(CH_2)_{x2} - \frac{R^4}{C} - (CH_2)_{z2} - \right]}$$

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and second mer units having the structure

$$(III) \qquad \frac{\left(\text{CH}_{2} \right)_{y3} - \left(\text{CH}_{2} \right)_{y3} - \left(\text{CH}_{2} \right)_{z3}}{\left(\text{CH}_{2} \right)_{z3}} = \frac{\left(\text{CH}_{2} \right)_{z3}$$

wherein:

15 R^4 and R^6 are independently selected from the group consisting of hydrogen, lower alkyl, lower alkoxy, $-COOR^8$ and $-(CH_2)_{n1}-O-R^8$ in which R^8 is lower alkyl or fluorinated lower alkyl and n1 is an integer in the range of 1 to 6 inclusive;

 R^5 is $-(CH_2)_{x4}(OCH_2CH_2)_{y4}(OCF_2CF_2)_{z4}SO_3M$ in which M is an alkali metal;

 $\rm R^7$ is $-(\rm CH_2)_{x5}(\rm OCH_2CH_2)_{y5}\rm OCH_3$ or $-\rm COOR^9$ in which $\rm R^9$ is lower alkyl or fluorinated lower alkyl; and

x2, x3, x4, x5, y2, y3, y4, y5, x2, z3 and z4 may be the same or different and are integers in the range of 1 to 100 inclusive.

18. The copolymer of claim 17 wherein M is lithium.

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19. The copolymer of claim 18 having a weight average molecular weight $\overline{M}_{\rm w}$ in the range of approximately 10,000 to 3,000,000.

20. The copolymer of claim 19 having a weight average molecular weight \overline{M}_{w} in the range of approximately 100,000 to 1,000,000.

- 21. The copolymer of claim 17 wherein R^4 and R^6 are independently selected from the group consisting of hydrogen and lower alkyl, and R^7 is $-(CH_2)_{x4}(OCH_2CH_2)_{y4}OCH_3$.
- 10 22. The copolymer of claim 17 wherein R⁴ and R⁷ may be the same or different, and are selected from the group consisting of moieties having the structure—COOR⁸, and R⁶ is hydrogen or lower alkyl.
- 23. A conductive composition having enhanced ambient temperature conductivity comprising:
 - (a) a single-ion conducting polymer having the structure

(I) $\frac{\mathbb{R}^2}{\left[\begin{array}{c} \mathbb{R}^2 \\ \mathbb{S}i \\ \mathbb{R}^1 \end{array}\right]^n}$

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wherein:

 ${\ensuremath{R}}^1$ and ${\ensuremath{R}}^2$ are individually selected from the group consisting of moieties having the structure

-(CH₂)_{x1}(OCH₂CH₂)_{y1}(OCF₂CF₂)_{z1}SO₂R³

in which R^3 is -OM, $-N(M)SO_2CF_3$ or $-C(M)(SO_2CF_3)_2$ and M is an alkali metal, or wherein one of R_1 and R_2 has the structure

$$-(CH_2)_{x1}(OCH_2CH_2)_{y1}(OCF_2CF_2)_{z1}SO_2R^3$$

and the other is selected from the group consisting of hydrogen, lower alkyl, fluorinated lower alkyl, lower alkenyl, fluorinated lower alkenyl, aryl and aralkylene, x1 and z1 may be the same or different and are integers in the range of 1 to 100 inclusive, y1 is an integer in the range of 0 to 100 inclusive, and n is an integer indicating the number of mer units in the polymer; and

- (b) a plasticizer selected from the group consisting of lower alkyl carbonates, glymes and cyclic ethers, in an amount effective to enhance the ionic conductivity of the copolymer.
- 15 24. A conductive composition having enhanced ambient temperature conductivity comprising:
 - (a) a single-ion conducting polymer electrolyte in the form of a polymer containing first mer units having the structure

(II)
$$\frac{1}{\left[(O)_{x2} - (CH_2)_{y2} - \frac{R^4}{C} - (CH_2)_{z2} \right]}$$

and second mer units having the structure

30 (III)
$$\frac{\left[(CH_2)_{y3} - \frac{R^6}{c} - (CH_2)_{z3} \right]}{\left[R^7 \right]}$$

wherein:

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 R^4 and R^6 are independently selected from the group consisting of hydrogen, lower alkyl, lower alkoxy, $-COOR^8$ and $-(CH_2)_{n1}-O-R^8$ in which R^8 is lower alkyl or fluorinated lower alkyl and n1 is an integer in the range of 1 to 6 inclusive;

 $\rm R^5$ is $-(\rm CH_2)_{\,x4}(\rm OCH_2CH_2)_{\,y4}(\rm OCF_2CF_2)_{\,z4}SO_3M$ in which M is an alkali metal;

 $\rm R^7$ is $-(\rm CH_2)_{x5}(\rm OCH_2CH_2)_{y5}\rm OCH_3$ or $-\rm COOR^9$ in which $\rm R^9$ is lower alkyl or fluorinated lower alkyl; and

- x2, x3, x4, x5, y2, y3, y4, y5, x2, z3 and z4 may be the same or different and are integers in the range of 1 to 100 inclusive; and
- (b) a plasticizer selected from the group consisting of lower alkyl carbonates, glymes and cyclic ethers, in an amount effective to enhance the ionic conductivity of the copolymer.
 - 25. The conductive composition of claim 23, in the form of a film.
 - 26. The conductive composition of claim 24, in the form of a film.
- 27. A single-ion conducting gel electrolyte film composition having enhanced ambient temperature conductivity and mechanical strength, comprising:
 - (a) the polymer of claim 1;
 - (b) a plasticizer selected from the group consisting of linear carbonates, glymes, cyclic ethers, and combinations thereof; and
 - (c) an amount of a strengthening material effective to enhance the mechanical strength of the film composition.

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28. A single-ion conducting gel electrolyte film composition having enhanced ambient temperature conductivity and mechanical strength, comprising:

(a) the polymer of claim 17;

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- (b) a plasticizer selected from the group consisting of linear carbonates, glymes, cyclic ethers, and combinations thereof; and
- (c) an amount of a strengthening materialeffective to enhance the mechanical strength of the filmcomposition.
 - 29. The composition of claim 27, wherein the strengthening material is polyvinylidene fluoride.
- 30. The composition of claim 28, wherein the strengthening material is polyvinylidene fluoride.
 - 31. A method of manufacturing a single-ion conducting gel electrolyte film having a selected film thickness, comprising the steps of:
 - (a) forming a gel electrolyte composition by combining (i) the single-ion conducting polymer of claim 1 with (ii) an amount of plasticizer effective to enhance the ionic conductivity of said polymer and (iii) an amount of a strengthening material effective to enhance mechanical strength; (b) heating the resulting composition at a temperature and for a time effective to form a fluid solution; (c) pressing the fluid solution between two solid substrates to form a film of a desired thickness; (d) cooling the film; and (e) releasing the film from the solid substrates.
 - 32. A method of manufacturing a single-ion conducting gel electrolyte film having a selected film thickness, comprising the steps of:

(a) forming a gel electrolyte composition by combining (i) the copolymer of claim 1 with (ii) an amount of plasticizer effective to enhance the ionic conductivity of said polymer and (iii) an amount of a strengthening material effective to enhance mechanical strength; (b) heating the resulting composition at a temperature and for a time effective to form a fluid solution; (c) pressing the fluid solution between two solid substrates to form a film; (d) cooling the film; and (e) releasing the film from the solid substrates.

- 33. The method of claim 31, wherein the strengthening material is polyvinylidene fluoride.
- 15 34. The method of claim 32, wherein the strengthening material is polyvinylidene fluoride.
- 35. A solid-state battery comprising a positive electrode, a negative electrode and the single-ion conducting polymer electrolyte of claim 1.
 - 36. A solid-state battery comprising a positive electrode, a negative electrode and the single-ion conducting polymer electrolyte of claim 17.

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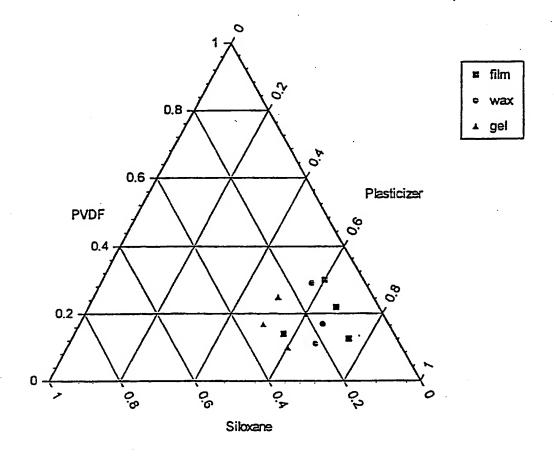


FIG. 1

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H01M 6/18, H01B 1/12, C08G 77/46, 65/22

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333

(71) Applicant:

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Redwood City, CA 94061 (US). VENTURA, Susanna, C.; 424 Becker Lane, Los Altos, CA 94022 (US).

(74) Agents: REED, Dianne, E.; Reed & Robins, Suite 200, 285 Hamilton Avenue, Palo Alto, CA 94301 (US) et al.

(81) Designated States: CA, JP, European patent (AT, BE, CH, DE, DK, ES, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE).

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With international search report.

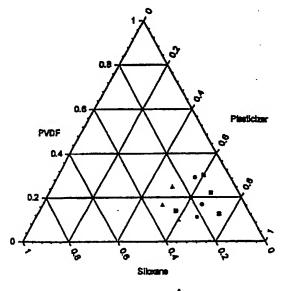
Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of

(88) Date of publication of the international search report: 12 September 1996 (12.09.96)

(54) Title: SINGLE-ION CONDUCTING SOLID POLYMER ELECTROLYTES

(57) Abstract

Novel single-ion conducting polymer electrolytes (SPEs) are provided. A first group of polymers are polysiloxanes substituted with fluorinated poly(alkylene oxide) side chains having associated ionic species, while a second group are copolymers containing mer units having structures (II) and (III), in which R4 through R8, x2, x3, y2, y3, z2 and z3 are as defined herein. Also provided are conductive compositions containing these novel polymers, particularly film compositions, and batteries formulated with such films.



$$\frac{1}{\left[\begin{array}{cccc} (0)_{23} - (CH_2)_{y3} - \frac{1}{c} - (CH_2)_{63} \\ \frac{1}{R^7} \end{array}\right]} \quad \text{(III)}$$

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According (to International Patent Classification (IPC) or to both national cla	seification and IPC	
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	see page 2203		
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	GB,A,2 119 162 (ATOMIC ENERGY AUTHORITY UK) 9 November 1983 see column 1, line 52 - line 58; claim 1	1,2
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Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of amendments.

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(72) Inventors: NARANG, Subhash, C.; 514 Rutherford Avenue, Redwood City, CA 94061 (US). VENTURA, Susanna, C.; 424 Becker Lane, Los Altos, CA 94022 (US).

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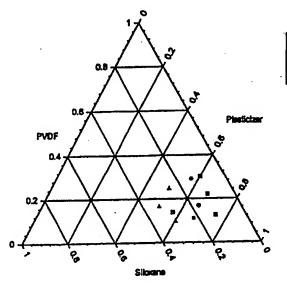
(81) Designated States: CA, JP, European patent (AT, BE, CH, DE,

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$$\frac{1}{(0)_{13}} - (CH_2)_{y3} - \frac{1}{6} - (CH_2)_{13} - \frac{1}{10}$$
 (III)

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Inten aal Application No
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A. CLASSIFICATION OF SUBJECT MATTER IPC 6 H01M6/18 H01B1/12 C08G77/46 C08G65/22 According to International Patent Classification (IPC) or to both national classification and IPC B. FIELDS SEARCHED Minimum documentation searched (classification system followed by classification symbols) IPC 6 HOIM HOIB COSG Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched Electronic data base consulted during the international search (name of data base and, where practical, search terms used) C. DOCUMENTS CONSIDERED TO BE RELEVANT Relevant to claim No. Citation of document, with indication, where appropriate, of the relevant passages Category * 1,2,5,6, WO,A,92 02571 (STANFORD RES INST INT) 20 A February 1992 see page 1, line 10 - line 13 see page 10, line 15 - line 18 see page 3, line 1 - line 21 1.2 MACROMOLECULES, A. vol. 26, no. 9, 26 April 1993, WASHINGTON US. pages 2202-2208, XP000362307 GUANG-BIN ZHOU ET AL: "SOLVENT-FREE CATION-CONDUCTING POLYSILOXANE ELECTROLYTES WITH PENDANT OLIGO (OXYETHYLENE) AND SULFONATE GROUPS" Scheme 1 see page 2203 -/--Patent family members are listed in annex. IX I Further documents are listed in the continuation of box C. X "T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention Special categories of cited documents: "A" document defining the general state of the art which is not considered to be of particular relevance "X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone "E" earlier document but published on or after the international filing date "Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such document, such combination being obvious to a person skilled in the art. "L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified) O* document referring to an oral disclosure, use, exhibition or other means document published prior to the international filing date but later than the priority date claimed "&" document member of the same patent family Date of mailing of the international search report Date of the actual completion of the international search 22.07.96 11 July 1996 Authorized officer Name and mailing address of the ISA European Patent Office, P.B. 5818 Patentiaan 2 NL - 2220 HV Rijwejk Td. (+31-70) 340-2040, Tx. 31 651 spo ni, Pate (+31-70) 340-3016 Andrews. M

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